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Drastic effect of shearing on graphite microtexture: attention and application to Earth science



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Abstract

The microtexture of graphite exposed on the polished surface was studied using confocal laser scanning microscopy, laser Raman spectroscopy, and focused ion beam–transmission electron microscopy (FIB–TEM) to elucidate the effect on surface condition and crystallinity of graphite by polishing process. The polished surface of the graphite was divided into a flat part with no irregularities and a grooved band with a width of < 1 μ m and a depth of < 100 nm. Raman analyses revealed that the original structure of the graphite covered by the host mineral was a well-ordered graphite, whereas the polished graphite at the surface had a reduced crystallinity, particularly in the flat part of the sample. Based on scanning TEM observations of an ultra-thin FIB section, fractures that developed during sample preparation were concentrated in the region extending from the surface to a depth of 1 μ m. Furthermore, graphite sheets were peeled away by shearing, with scraped graphite sheets filling in the gap. Our results demonstrate that the original microtexture of graphite was easily deformed by shearing during polishing, and careful attention should be paid to sample preparation. In addition, we also need to pay more attention to the effects of natural shearing such as faulting on the graphite or sheet-like minerals.

Keywords: Graphite, Shearing, Microtexture, Confocal laser scanning microscope, Raman spectroscopy, FIB–TEM, Fault rock, Sheet-like mineral

Introduction

Carbon is a widely distributed element on the Earth, occurring from the upper atmosphere to the deep interior of the Earth. One of the forms of carbon present in rocks is carbonaceous material (CM), which is ubiquitous within sedimentary and metasedimentary rocks. Carbon has been increasingly used in Earth sciences since the Raman CM geothermometer for regional metamorphic rocks was proposed by Beyssac et al. (2002). More recent studies have extended the applicable range of Raman CM geothermometers to contact and low-grade metamorphic rocks including diagenetic sedimentary rocks, covering a temperature range from ~ 150 to 650 °C (e.g., Aoya et al. 2010; Lahfid et al. 2010; Kouketsu et al. 2014; Mori et al. 2017). In addition, Raman spectroscopy is increasingly being applied to studies of CM in meteorites and fault rocks

CM exposed on the surface of samples had been damaged by polishing, resulting in a decrease in the crystallinity

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⁽e.g., Busemann et al. 2007; Muirhead et al. 2012; Furuichi et al. 2015; Hirono et al. 2015; Homma et al. 2015). Because structure of CM reflects the heat effect, several studies have succeeded in quantitative determination of the petrologic types of chondrites (Quirico et al. 2003; Bonal et al. 2006). Also utilizing these properties, Busemann et al. (2007) and Homma et al. (2015) proposed the thermometer for meteorites. Several studies successfully detected the shear heating recorded in the CM of the fault rocks and discussed the properties of earthquake slips such as displacement rate and coefficient of friction (Mori et al. 2015; Hirono et al. 2015). On the other hand, Nakamura et al. (2015) and Kouketsu et al. (2017) reported the mechanical damage on CM in the fault rocks, and they pointed out that not only the evolution of CM crystallinity by shear heating but also the destruction of crystal structure by shearing should be taken into consideration in the fault rocks.

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of CM, which adversely affects the accuracy of temperature estimates (Beyssac et al. 2003; Kirilova et al. 2018). However, the large grained CM in several samples is easy to be exposed on the surface of a thin section and it is difficult to find CM that is completely buried within the host minerals. Alternatively, there are some studies carrying out the Raman spectroscopic analysis on polished CM that has been separated by acid treatment and embedded in resin (Lünsdorf 2016). In such cases, it is necessary to consider the effect on the CM of polishing during sample preparation, which is poorly understood.

In this study, we examined graphite exposed on the surface of a thin section during sample preparation by using confocal laser scanning microscopy (CLSM), laser Raman spectroscopy, and focused ion beam—transmission electron microscopy (FIB—TEM), to investigate the effects of shearing by polishing on graphite microtexture. Using these data, we detail the risks of applying the Raman CM geothermometer to exposed CM and consider the effect of shearing of CM by faulting, which is of the same mechanism as how it operates during polishing. We also consider the effect of shearing on other sheet-like minerals.

Sample

We used a graphite grain in sample S26 that was collected from granulite facies region of Mogok metamorphic belt, Myanmar, which is located along the western margin of the Shan-Tai block. The sample S26 is a garnet-biotite-sillimanite gneiss, and its petrological and mineralogical characteristics were documented by Maw Maw Win et al. (2017). It is composed mainly of garnet, biotite, plagioclase, quartz, and sillimanite, with minor graphite, rutile, ilmenite, apatite, monazite, and zircon. The temperature conditions at peak metamorphic stage estimated by garnet-biotite geothermometer (Bhattacharya et al. 1992; Holdaway 2000) and Zr-in-rutile geothermometer (Tomkins et al. 2007) were in excess of 800 °C (Maw Maw Win et al. 2016, 2017).

Analytical methods

The rock sample was cut with a diamond-tipped saw and cut out of $\sim 4\,\mathrm{cm}$ length and 2.5 cm width shape. The one side of the rock chip was polished using a grinder with diamond discs of 20 cm diameter (#120, #220, #600, #1200, and #2000). The rotational speed was set to 100 rpm, and each polishing was kept at 3 min. The chip was attached to the stainless holder, whose weight is 455 g, with a double-sided tape. The polished rock chip was stuck to the slide glass using petropoxy and cut with a diamond-tipped saw. The thin section (\sim 30 μ m thickness) was polished by hand using silicon carbide and aluminum oxide (#120, #400, #600, #1200, and #3000); a final polishing step used 1 μ m followed by

 $0.25\,\mu m$ diamond pastes. Changes in the degree of crystallinity of graphite at each polishing step were evaluated by Raman spectroscopy using the chip sample. The thin section was analyzed by CLSM, Raman spectrometry to check the surface form and crystallinity of the graphite. After these analyses, an ultra-thin section was cut out using FIB and then the cross section of the graphite was observed using TEM. Following these analyses, chemo-mechanical polishing of the sample using colloidal silica was also carried out for 30 h, after which the crystallinity of graphite was examined by Raman spectroscopic analysis.

The surface morphology was observed using a CLSM (LEXT OSL3100; Olympus, Tokyo, Japan) at the Geochemical Research Center at the University of Tokyo, Tokyo, Japan. In the confocal mode, an image is generated from the reflection of light on the focal plane where the light is the most intense. The source of the light is a laser diode emitting a light beam with a wavelength (λ) of 408 nm. The spatial resolution is 120 nm in the XY direction and 70 nm in the Z direction (Kaplonek and Nadolny 2012).

The crystallinity of graphite was measured using a Nicolet Almega XR laser Raman spectroscope (Thermo Fisher Scientific) at the Petrology and Mineralogy Laboratory at Nagoya University, Nagoya, Japan. A 532 nm Nd-YAG laser passed through a confocal microscope (BX51; Olympus, Tokyo, Japan) with a ×100 objective (Olympus Mplan-BD 100X, NA = 0.90) was used for measurement. The laser power at the sample surface was set to ~2 mW to avoid irradiation damage to the graphite surface. The scattered light was collected using a backscattered geometry with a 25 µm pinhole and a holographic notch filter, dispersed using a 2400 lines/ mm grating, and analyzed by a Peltier cooled CCD detector comprising 256 × 1024 pixels (Andor Technology, Belfast, Northern Ireland). The acquisition time of the CM spectra was 30 s, the spatial resolution was $\sim 1 \,\mu\text{m}$, and the wavenumber resolution was $\sim 1 \text{ cm}^{-1}$. Raman spectra of well-ordered graphite show a single peak at around 1580 cm⁻¹ (G-band), whereas additional peaks due to the reduction of crystallite size in the direction of the graphite sheet caused by defects or a disordered structure appear at around 1350 cm⁻¹ (D1-band) and 1610 cm⁻¹ (D2-band) (Tuinstra and Koenig 1970). In this study, measured Raman spectra were decomposed using PeakFit 4.12 software (SeaSolve Software) with a pseudo-Voigt function (Gaussian- Lorentzian Sum) by subtracting a linear baseline in the spectral range of 1000–1800 cm⁻¹. The parameters of R1 which is the intensity ratio of D1- and G-bands (Tuinstra and Koenig 1970) and R2 which is the area ratio of D1-, D2-, and G-bands (Beyssac et al. 2002) were used to evaluate the crystallinity of graphite.

$$R1 = (D1/G)_{Intensity} \tag{1}$$

$$R2 = D1/(G + D1 + D2)_{Area}$$
 (2)

The values of R1 and R2 become smaller with increasing graphite crystallinity. The value of R1 correlates with the crystal size of lateral extent of carbon sheet (Tuinstra and Koenig 1970; Ferrari and Robertson 2000). The upper limit of the temperature determined from R2 for graphite in regional metamorphic rocks is 650 °C, at which temperature the D-bands disappear and only the G-band remains.

Following the CLSM and Raman spectroscopic measurements, an ultra-thin section of the sample was made using the FIB system (Helios NanoLab G3 CX, Thermo Fisher Scientific) at Kyoto University, Kyoto, Japan. A predefined area (~30 µm²) was coated with platinum (Pt) and the surroundings cut to a depth of $\sim 10-15 \,\mu m$ using a gallium (Ga) ion beam. The resulting foil was picked up using a tungsten (W) probe and mounted on a TEM grid. The extracted samples were thinned to 100-300 nm using a Ga ion beam employing a voltage of 30 kV and beam currents ranging from 0.1 to 3 nA and then using a voltage of 5 kV and a beam current of 16 pA for the final processing. The sample was studied using a TEM equipped with an energy-dispersive spectroscopy (EDS) system operating at 200 kV at Kyoto University (JEM-2100F, JEOL) and Nagoya University (JEM-2100 with a Dry SD60GV detector, JEOL). TEM images were captured using a CCD camera (Orius200D or ES500W Erlangshen, Gatan).

Results

Morphology of polished graphite

The graphite grains show elongated shape with lengths up to several millimeters (Fig. 1). Due to their large grain size, almost all graphite grains are exposed on the surface of the thin section. Several graphite grains exposed on the surface shows unidirectional streaks parallel to the elongate direction of the graphite under a reflection microscope (Fig. 1a, b, d). CLMS analysis revealed that the surface morphology of the streak part corresponds to the groove shape (Fig. 2). The width of the groove was 1 μm or less, and the maximum depth was 100 nm or less. The other part shows a flat shape without remarkable irregularities. In the following description, we subdivide the graphite surface into the grooved band and the flat part.

Crystallinity degree of graphite

Change of the graphite crystallinity by each polishing step

We evaluated the change of crystallinity of graphite in a rock chip at each polishing step by Raman spectroscopy. The crystallinity of the fracture surface of the graphite exposed on the rock specimen was that of a well-ordered graphite in which only the G-band was observed without any D-bands (Fig. 3a). D1-band was slightly observed at some points after cutting by a diamond-tipped saw, and the degree of crystallinity is decreased (Fig. 3b). The surface form is rugged and mussed under the reflection microscope (Fig. 3c, f). After the #120 polishing, the crystallinity once recovers but thereafter lowers as a whole although a slight recovery is observed in some steps (Fig. 3b). The surface form becomes nearly flat after the #1200 polishing (Fig. 3d, g). After grinding by 1 μ m diamond paste for 10 min, streaks began to be seen under the reflection microscope (Fig. 3h).

Relation of the surface morphology and crystallinity

The Raman spectra of both grooved band and the flat part show G-, D1-, and D2-bands (Fig. 4b). The intensity of the D-bands in the flat part was stronger than that in the grooved band, suggesting that the crystallinity in the flat part was lower than that in the grooved band (Tuinstra and Koenig 1970; Ferrari and Robertson 2000). For comparison, we measured the graphite grain that was buried within the host K-feldspar, for which only the G-band was observed (Fig. 4c). Step scan measurements of 1 µm interval across the boundary between the grooved band and the flat part show that the R1 and R2 values are lower in the grooved bands (0.4-0.6 and 0.4-0.5, respectively) than in flat parts on both sides (0.6–0.8 and 0.5–0.6, respectively; Fig. 4d). The R1 and R2 values of the grooved band are correlated to its width and lower in a wider grooved band. Additional chemo-mechanical polishing using colloidal silica causes little change in the crystallinity in the flat part. On the other hand, the increases of R1 and R2 values in the grooved part suggest reduction of the crystallinity of the graphite (Fig. 5).

Microtexture of polished graphite

Annular dark-field scanning TEM (ADF-STEM) images show that many fractures are developed near the polished surface under the flat part of the sample (the highly fractured zone in Fig. 6a). These fractures reach a depth of $\sim 1 \,\mu m$. In addition, some cleavage planes extend to the bottom of the ultra-thin section of the graphite to a depth of more than 6 µm. Kinks are developed at a depth of $\sim 5 \,\mu m$ under the flat part, and the density of cleavages decrease in the lower part of the sample. In contrast, the grooved band shows different textures in that noticeable fractures are not visible, but bent or folded structures of graphite sheets continue to a depth of $\sim 5 \,\mu m$. The STEM-EDS analysis shows only carbon within the mapped area; other elements such as Si, Al, and O were not detected (Additional file 1: Figure S1).

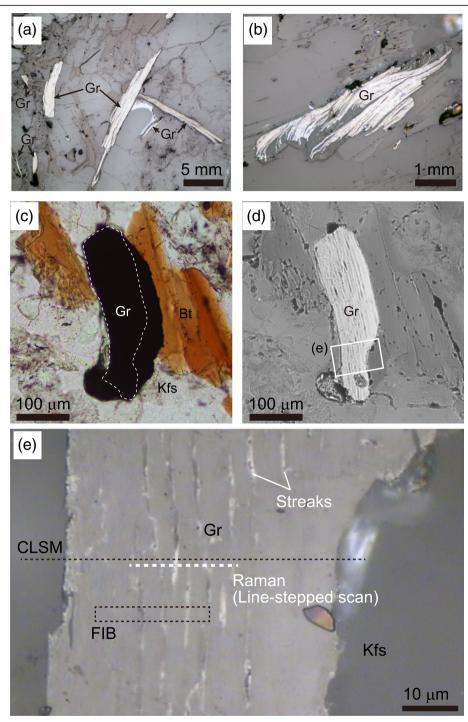


Fig. 1 Photomicrograph of the analyzed graphite. **a, b** Graphite grains under reflected light. **c** Studied graphite grain under transmitted light. The area exposed on the surface of graphite is surrounded by a white dotted line. Minerals surrounding the graphite are biotite and K-feldspar. **d** Studied graphite grain under reflected light. The area shown in (**e**) is indicated by a white square. **e** Magnified reflective microscope image of studied graphite. Streaks (grooves) extending in one direction can be seen on the graphite surface. The black dashed line and white dashed line indicate the place where the CLSM and the line-stepped scan by Raman spectroscopy were preformed, respectively. The area cut out using the FIB is enclosed by a black square. Abbreviations are Bt, biotite; Gr, graphite; Kfs, K-feldspar

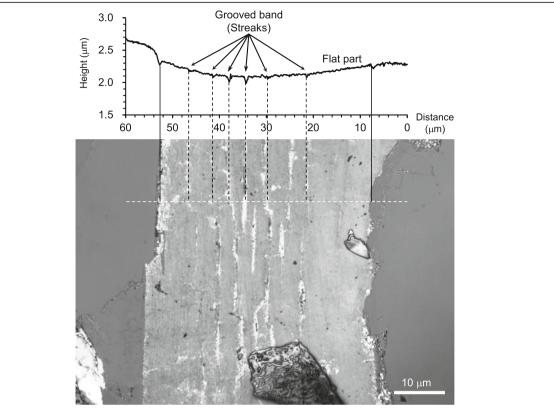


Fig. 2 Depth profile through the polished graphite with the CLSM image focused at a certain depth. The confirmed locations of grooves are indicated by thick dotted lines. The edges of graphite are indicated by solid lines

Selected area electron diffraction (SAED) analysis of the bottom part of the sample, 5 µm below the surface of the flat part, shows a clear diffraction pattern and could index as graphite along the [010] zone axes (Fig. 6c). The clear spots indicate that the crystal structure of graphite is well ordered. The SAED pattern in the $1.5\,\mu m$ below the surface of the flat part also shows clear diffraction patterns, but streaking of the diffraction spots and weak superlattice reflections along the c^* direction are observed, suggesting sheet-stacking errors (stacking fault; Fig. 6d). In the highly fractured zone beneath the flat part, the SAED pattern is unclear (Fig. 6e), implying grain size reduction and/or disordering of crystal. Beneath the grooved band, the orientation of graphite sheets varies, although in many case the 00 l spots can be identified (Fig. 6f, g).

Discussions

Changes in graphite crystallinity during sample preparation

The change of Raman spectra and its R2 values indicate that the crystallinity of graphite is easily decreased at each sample preparation step from cutting by the diamond-tipped saw to polishing with fine-grained diamond pastes. The reflection microscopic image of the surface condition of graphite after cutting with the

diamond-tipped saw is messy, and the crystal structure may have been affected by the violent deformation with the blade rotating at a high speed. On the other hand, the crystallinity was once recovered in #120 polish step, and the surface subjected to a destructive deformation was likely removed by rough polishing. However, the subsequent polishes tend to degrade the crystallinity, and the crystal structure is disturbed by polishing with diamond or colloidal silica finally. Looking at the graphite cross section of the STEM image, cracks of various lengths were observed from $\sim 1\,\mu\mathrm{m}$ length at highly fracture zones to $>\sim 10\,\mu\mathrm{m}$ cleavage continuing to the bottom of the ultra-thin section. These cracks in the graphite would occur at each cutting and polishing step corresponding to the grain size of abrasive powders.

Chemo-mechanical polishing with colloidal silica resulted in little change to the crystallinity of the graphite in the flat part (Fig. 5). This is because the fractures reached a depth of $\sim 1\,\mu m$ and could not be erased by the several nanometers of polishing that results from this chemo-mechanical process. Furthermore, as the crystallinity of graphite was reduced in the grooved band by the colloidal silica polishing, it is possible that vibrations due to chemo-mechanical polishing induced further fractures to form in the graphite sheet.

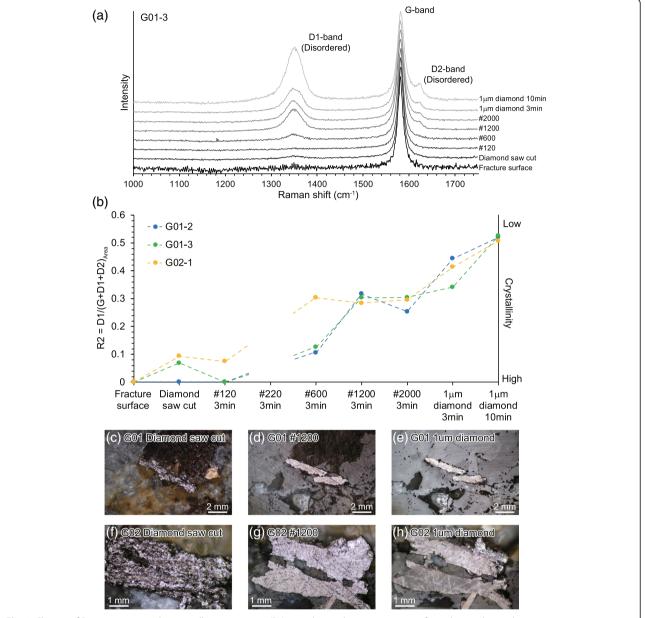


Fig. 3 Change of Raman spectra and its crystallinity parameter (R2) at each sample preparation step from diamond-tipped saw cutting to 1 μm diamond paste on three spots (G01-2, G01-3, G02-1). **a** Change of Raman spectra at each sample preparation step at the spot of G01-3. **b** Change of crystallinity parameter (R2) at each step except #220. Raman measurement after #220 was skipped, and its R2 data are blank **c-h** Surface shape change under reflected microscope after cutting by a diamond-tipped saw (**c, f**), after 3 min by #1200 polishing (**d, g**), and after 10 min by diamond paste polishing (**e, h**) of grains G01 and G02. Please note that the rough polishing steps scrap the surface greatly and not always measure exactly the same point as in previous polishing steps

Modification process of graphite microtexture

STEM observation revealed that the damage on the crystal surface of the flat part by polishing process continues to a depth of $\sim 1\,\mu m$ forming a highly fractured zone. Sheet-like graphite structure continues to a depth of more than $6\,\mu m$ beneath the flat part. The cleavages in graphite sheets under the flat part are developed along the (001) plane, and graphite sheets are readily peeled away from that plane. Bending or folding

of graphite sheets were observed beneath the grooved band continued to a depth of $\sim 5~\mu m.$ This complicated microtexture is considered to be formed by the sample preparation process, not the original structure of graphite.

Based on the idea that the cleavages and fractures occur at each sample preparation step corresponding to the grain size of abrasive powders, Fig. 6a is considered to have been formed in the following steps.

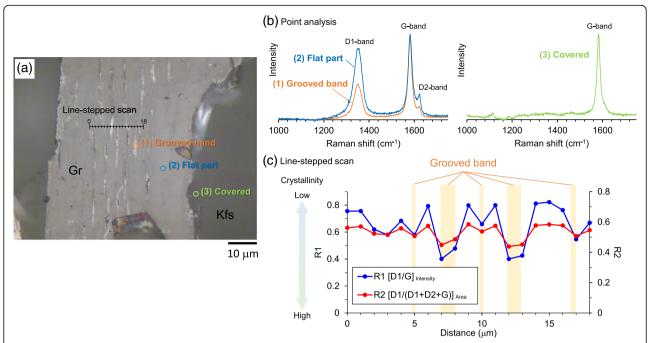


Fig. 4 Raman spectroscopic analysis of graphite. **a** Reflected light image of graphite showing the locations of point analyses and the line-stepped scans. **b** Raman spectra of point analyses of graphite for (1) the grooved band, (2) the flat part, and (3) the covered part. **c** Line-stepped scan of the graphite surface. The blue data indicate the value of R1 and the red data indicate the R2 values. The yellow shading indicates the locations of grooved bands

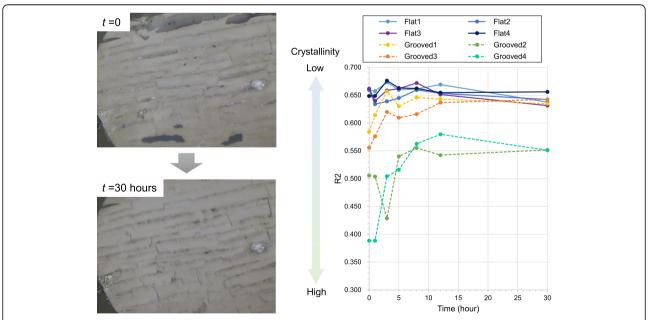


Fig. 5 Change of the graphite crystallinity by chemo-mechanical polishing using colloidal silica. Chemo-mechanical polishing was operated using VibroMet™ 2 Vibratory Polisher. The photographs on the left show the reflection microscopic images of the graphite before and 30 h after the chemo-mechanical polishing. The graph on the right shows the changes of graphite crystallinity (=R2) with time. The data in the flat part are indicated by solid lines, and the data of grooved band are indicated by dotted lines

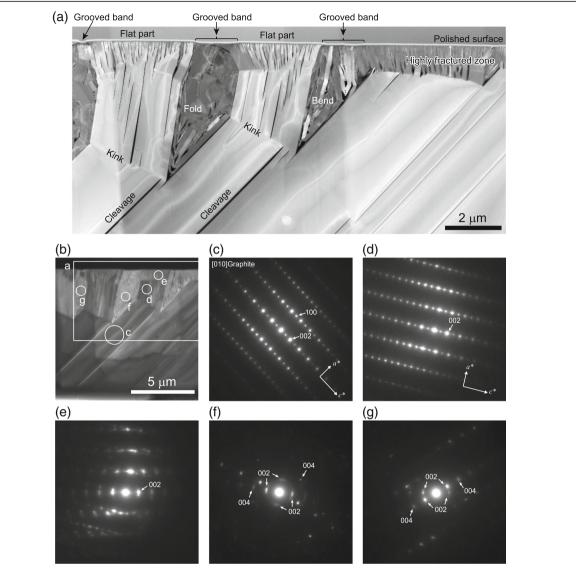


Fig. 6 a STEM image of the ultra-thin section of graphite cut by FIB. The top of the STEM image is the polished surface, which is covered by a coating of platinum and carbon. **b** Bright field (BF) TEM image of an ultra-thin section of graphite as cut by FIB. The white circles show the location of spots of the SAED patterns within the measured areas in (**c**) to (**g**). **c** The SAED pattern of the bottom part of the graphite sheet under the flat part shows clear 100 and 001 spots. **d** The SAED pattern of the area ~ 1.5 μm below the flat part shows clear 100 and 001 spots. **e** The SAED pattern of the highly fractured zone in the flat part exhibits 100 and 001 spots. **f** The SAED pattern of the area under the grooved band showing weak 002 spots. **g** The SAED pattern of the area under a different grooved band indicate weak 002 spots

- (1) Well-ordered graphite was crystallized within the rock under the high-temperature metamorphic condition (Fig. 7a).
- (2) The surface is disturbed after cutting by diamond-tipped saw. Cleavages developed within the graphite sheets (Fig. 7b).
- (3) The sheets of ordered graphite are peeled in the weakest direction due to shearing during polishing. As a result, gaps are generated between the peeled graphite sheets, and kinks are also developed. The peeled-off graphite sheets fill the gap. (Fig. 7c).
- (4) The peeled-off graphite sheets continued to fill the gap. The fractures under the flat part develop further polishing by finer grained abrasive powders (Fig. 7d).

In the analyzed sample, S26 was recrystallized under the granulite facies condition of temperature over 800 °C (Maw Maw Win et al. 2016, 2017), and the Raman spectrum of the graphite before sample preparation or covered by host mineral (K-feldspar) shows only a G-band but no D-bands (Figs. 3a and 4b). These data indicate that the original structure of the graphite grain

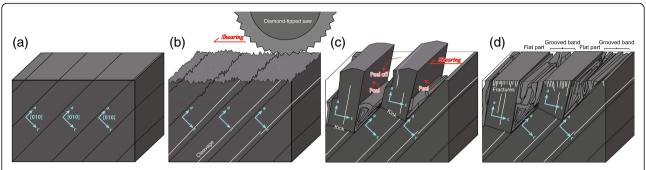


Fig. 7 Schematic diagram showing deformation of the crystal structure of polished graphite. **a** Original graphite structure prior to sample preparation. **b** Graphite structure after cutting by a diamond-tipped saw. The surface condition is disturbed, and cleavages developed within the graphite sheets. **c** Graphite sheets are peeled by shearing during polishing, and a gap is formed between them. The kink is formed by bending of the graphite sheet. The peeled graphite sheet further peels off, and fragments of the sheet fill the gap. **d** The final crystal structure of graphite is disturbed by polishing. Fractures are concentrated near the polished surface

was a well-ordered crystal. Although the crystallographic axes of the graphite are rotated by shear in the flat part, the a- and c-axes rotated within the same plane. This means that the graphite sheet is easiest to peel when shearing is applied in a direction closest to the c-axis direction on the surface of the thin section. On the other hand, in the grooved band, the c-axis maintains the same in-plane direction, while the a-axis has a variable orientation (Fig. 6f, g). Since the peeled-off graphite sheet also has a sheeted structure, the plane of the peeled graphite filling the gap may be easiest to fit along the gap plane that is sub-parallel to the original graphite plane.

The TEM and SAED analyses reveal that kink induces slip of graphite sheets and the formation of stacking faults. At the same time, defects within the graphite sheet are formed at the kink. Thus, the concentration of stacking defects increases notably above the large kink, but in-sheet ordering was almost unchanged except at the kink. In contrast, the highly fractured zone can be interpreted as a high concentration of very small kinks. Thus, in addition to the stacking faults, the in-sheet defects that affect the increase in D-band intensity of the Raman spectra are expected to be abundant in this zone.

Analysis by CLMS and Raman spectroscopy revealed that the crystallinity of the surface in the flat part was lowest and most strongly influenced by polishing during sample preparation. In contrast, the crystallinity of the grooved band was not as intensely disturbed as the flat part. This probably means that the grooved band was less affected by the shearing during polishing because of its concave shape.

Implications

Effect of shearing on graphite for Raman CM geothermometry

Beyssac et al. (2003) reported that the R2 parameter of graphite on polished surfaces is larger than that of buried graphite, especially for high-ordered graphite, and

advised not to use exposed graphite grains for Raman CM geothermometry. In this study, we also have confirmed that the crystallinity of graphite decreases with polishing. Since the Raman spectra of graphite covered by a host mineral show only a G-band (Fig. 4b), R2 is zero and the temperature is calculated to be $> 650^{\circ}$ C, whereas an R2 value of 0.5-0.6 in the flat part corresponds to temperatures of ~400 °C and an R2 of 0.4-0.5 in the grooved band reflects temperatures of ~ 450 °C based on the Raman CM geothermometer of Aoya et al. (2010). Based on these results, the estimated temperature of CM exposed on the surface by polishing is not reliable. To eliminate such uncertainties when applying the Raman CM geothermometer, it is necessary to carefully judge whether the graphite is buried within the host mineral by using reflected light microscopy.

Effect of shearing on graphite in fault rocks

Recent studies have tried to extract information on fault movement during earthquakes using CM grains (e.g., Kaneki and Hirono 2018). Since graphite has a very low shear strength, it is likely that the microtexture of graphite is readily modified by shearing associated with faulting. For example, Nakamura et al. (2015) reported the shear-induced amorphization of graphite in a fault zone in the Hidaka metamorphic belt. This evidence indicates shearing-related damage to graphite during fault movement. Kouketsu et al. (2017) also reported the signal of mechanical destruction of low-grade CM at Longmenshan thrust belt. On the other hand, Kaneki et al. (2018) reported the enhancement of organochemical reaction related to the aromatization of low-grade CM by the physical damage caused by shearing. In that case, the maturation of low-grade CM (coalification) is promoted by shearing. Several studies reported the shear-induced graphitization of amorphous carbon at high-velocity experiments (1-3 m/s) at high normal stress (2-25 MPa)

and pointed out the dynamic weakening of fault motion at a carbon-rich fault (Oohashi et al. 2011; Kuo et al. 2014).

In our sample preparation condition, the rotation velocity and normal stress is roughly calculated as 1 m/s and ~ 5 kPa. This condition is comparable to fast slip and very shallow (almost surface) faulting conditions. Some previous studies pointed out that the maturation of CM progress by shearing in some cases under the high normal stress condition. However, the degree of graphite crystallinity is lowered due to seismic shear at a high speed and close to the surface condition as in this study. We have to take into account that the deformation mechanism of graphite may change depending on the shearing condition such as normal stress and slip rate.

Effect of shearing on minerals

It has also been suggested that other sheet-like minerals, in addition to graphite, are deformed by polishing. Inoue and Kogure (2012) reported that crystals of phyllosilicates, muscovite, and chlorite are also bent by polishing process (to colloidal silica finally) and did not show sharp EBSD patterns. They carried out the ion milling (ion beam edging) and obtained sharp EBSD patterns with no crystal bending. In addition to muscovite and chlorite, the EBSD patterns of low-temperature serpentine such as lizardite and chrysotile have not been reported yet (c.f. Boudier et al. 2010; Bandli and Gunter 2013). One reason for the lack of EBSD pattern of low-temperature serpentine is discussed that the EBSD pattern would not be produced in the scrolled structure of the silicate sheets in chrysotile (Bandli and Gunter 2013). On the other hand, the other reason is presumed to be the effect of damage by mechanical polishing as with the graphite of this study.

The original structure of graphite was artificially changed during conventional polishing, and even during chemo-mechanical polishing using colloidal silica. To expose surfaces of graphite retaining their original microtexture, it is necessary to improve the polishing method, for example, by using a different technique such as ion milling (e.g., Sasaki et al. 2004; Inoue and Kogure 2012).

Conclusions

We analyzed graphite exposed on the surface of a sample by polishing using CLSM, laser Raman spectroscopy, and FIB–TEM measurements. The following results were obtained.

(1) CLSM measurement revealed that the polished graphite surface can be divided into a flat part and a grooved band (Fig. 2). The flat part is a flat plane without any noticeable irregularities, whereas the

- grooved band has a shape with a width of $<1\,\mu m$ and a depth of <100 nm.
- (2) In Raman spectroscopic analysis, both the flat and grooved bands show D-bands in addition to a G-band, whereas the graphite covered by the host mineral shows only a G-band (Fig. 4). This demonstrates that the crystallinity of graphite exposed on the surface was decreased by polishing. In particular, the D-band intensity in the flat part was stronger than that in the grooved band, and the flat part was more strongly affected by shearing than the grooved band.
- (3) FIB–TEM measurement revealed that fractures are developed within the graphite sheets that are concentrated in the region from the surface to a depth of $\sim 1~\mu m$ (the highly fractured zone; Fig. 6a). Based on the structures observed with STEM and the crystal orientations determined with TEM, we consider that the graphite sheets were peeled by shearing during polishing and that the scarped graphite sheets filled the gap produced between the peeled graphite sheets (Fig. 7).

The crystal structure of the graphite was deformed during all stages of sample preparation, from cutting of the sample through the various stages of polishing, including chemo-mechanical polishing using colloidal silica. These results indicate the need to consider the deformation of graphite microtexture arising from shearing, not only during sample preparation but also due to shearing caused by fault movement. In addition, it is necessary to devise a different method of sample preparation to obtain primary information on certain minerals, in particular for sheet minerals similar to graphite.

Additional file

Additional file 1: Figure S1. STEM–EDS mapping data. (a) STEM image of mapped area, (b) EDS spectrum of the entire mapped area. (c) to (f) shows element maps of carbon (C), silicon (Si), aluminum (Al), and oxygen (O), respectively. Only C was detected in the map area, and impurities such as Si, Al, and O were not detected. In the EDS spectrum, small peaks of Si, Fe, and Co peaks are derived from pole piece and/or TEM sample holder, Ga and Pt peaks were derived at the FIB process, and Cu peaks were derived from the sample holder. (PDF 133 kb)

Abbreviations

CLSM: Confocal laser scanning microscope; FIB: Focused ion beam; SAED: Selected area electron diffraction; TEM: Transmitted electron microscope

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Availability of data and materials

All original data are available from the corresponding author on request.

Authors' contributions

YK and ME proposed the project and conceived and designed the study. YK carried out the Raman spectroscopic analysis. AM, YI, and TT undertook the FIB–TEM analysis and analyzed the data. HK made the CLSM measurements. All authors discussed the results and interpreted the data. All authors read and approved the final manuscript.

Competing interests

The authors declare that they have no competing interests.

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